



Low-Cycle Fatigue and Dynamic Fracture in Gold Thin Films on SiN Supported Membranes

C. C. Hays, J. M. Newell, P. D. MacNeal, R. P. Ruiz, W. A. Holmes,
M. Yun, J. L. Mulder, T. C. Koch, J. J. Bock, and A. E. Lange
*Jet Propulsion Laboratory,
California Institute of Technology, Pasadena, CA*

Symposium Q: Degradation Processes in Nanostructured Materials

2005 MRS Fall Meeting
November 28 - December 2 | Boston, MA

Outline



- Introduction
- Experimental
 - description of thin-film structures
 - experimental methods
- Description of Device
- Mechanical response to vibration: substrate motion
 - closed form solutions
 - finite element modeling (FEM)
- Experimental Results
 - optical and scanning electron microscopy
- Discussion of Results
- Conclusion
- Acknowledgements

Introduction



- NASA/JPL uses latest cryogenic detectors to search for present-day signatures of the big-bang, and the following inflationary period ($t \approx 10^{38}$ sec)
- The Cosmic Microwave Background (CMB) has been measured by experiments such as COBE, BOOMERANG, DASI, and WMAP
- The next generation of missions will search for gravity-wave signatures on the CMB; i.e., polarization of the CMB
- Advanced cryogenic detectors that operate at 100 mK are required
- Rigorous tests are conducted prior to launch; e.g., a short list includes
 - Optical tests at low-T
 - Thermal Cycling
 - Vibration
- This talk focuses on the dynamic mechanical response and fatigue behavior in sub-micron thick Au-films deposited onto amorphous Si_xN_y substrates, with spider-web geometry, that were subjected to forced vibration (3-axis random vibration with 2 kHz roll-off frequency)

Experimental Methods



- Fabrication - thin film deposition on low stress, 1 μm thick silicon nitride substrates
- Vibration Test Spectrum
 - 3-axis random vibration with 2 kHz roll-off frequency
- Microscopy
 - Optical microscopy
 - Scanning electron microscopy



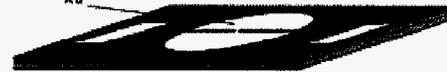
Fabrication Process

- Fabrication conducted in JPL's Micro Devices Laboratory (MDL).
- Au-metallized detectors are fabricated on thermally bonded SOI wafers with a (100) orientation
 - SOI wafers are prepared with a standard RCA cleaning
 - 1 μm layer of Si_3N_4 deposited on each side of the wafer using low-pressure chemical vapor deposition (LPCVD)
 - Si_3N_4 layer will ultimately form a large (~ 5 mm) free-standing structure, care must be taken in the deposition of the Si_3N_4 to ensure that the material is free of stress and pinholes
 - Ti-Au metal films are deposited using lift-off technique to form an IR-absorber layer
 - Au layer deposited on absorber chosen to give optimal infrared absorption, and is typically ~ 12 nm thick
 - Au layer forming the electrical lead layer is typically ~ 300 nm thick
 - reactive ion etching used to define the silicon nitride frame

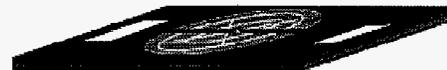
Fabrication Process Flow Diagram



(a) Deposition of Silicon Nitride on SOI wafer
Au



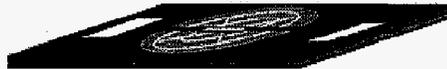
(b) Deposition of Au absorber, Lead, and Contact layers



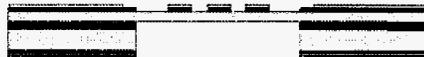
(c) Pattern Absorber and Si_3N_4 with photoresist



(d) Dry etch absorber and backside Si_3N_4



(e) In bump deposition and placing NTD Ga thermister chip



(f) Backside deep trench RIE and wet etching of buried oxide



(g) Final wet etching of top Si layer

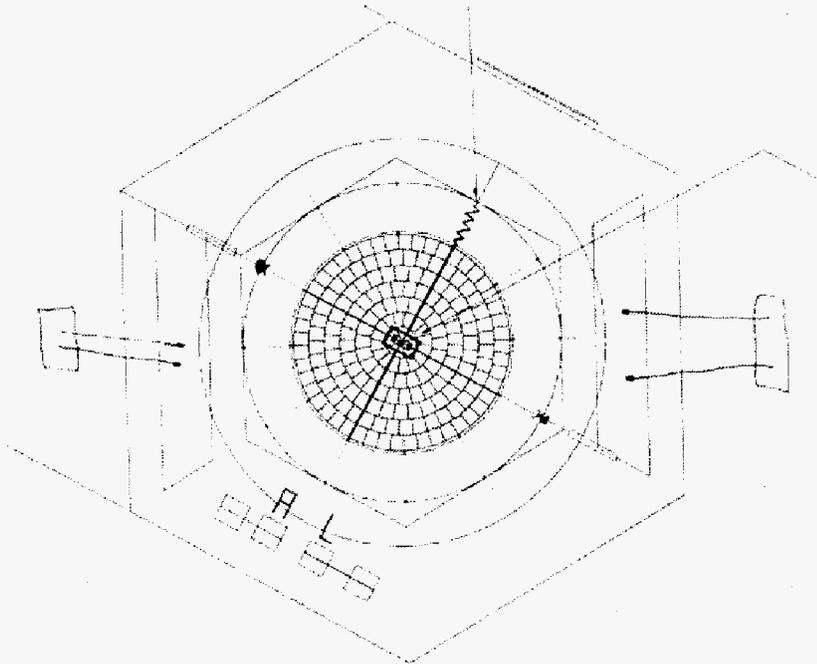


(h) Top view of the completed single detector

C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Description of Device



Web OD = 3.16 mm

Lead length = 4.66 mm

Au lead thickness = $0.3 \mu\text{m}$

Lead width = $5.3 \mu\text{m}$

Web width = $3.4 \mu\text{m}$

Web thickness = $1 \mu\text{m}$

Center chip size: $300 \mu\text{m} \times 100 \mu\text{m} \times 25 \mu\text{m}$

C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California



Vibration Test Spectrum

Frequency Range (Hz)	Spectral Density or Slope
20	.0323 g^2/Hz
20-75	+6 dB/octave
75-400	0.225 g^2/Hz
400-2000	-6. dB/octave
2000	.0182 g^2/Hz

Test duration: One minute per axis 12.3 G_{rms} overall

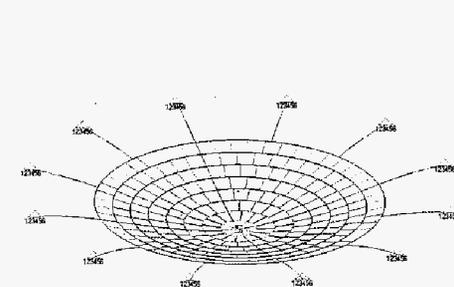
- Detectors mounted inside sealed carrier plates; hence the units under test are not visible during the test.
- 3-axis random vibration with 2 kHz roll-off frequency

Mechanical response to vibration



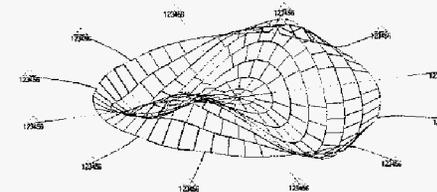
Images show results of FEM calculations for stress-free substrate and substrates with residual stress

FEM results agree (\approx) with closed form solution for tympani mode: $f_{\text{resonance}} \approx 1400$ Hz



Default Set Mode 1: 139.374 Hz
Deformed: 27102.1 Total Deformation
Color: Strain Energy Density

Fundamental Mode
No preload; 139.0 Hz



Default Set Mode 1: 2095.184 Hz
Deformed: 27296.1 Total Deformation
Color: Strain Energy Density

With preload; 2095.1 Hz

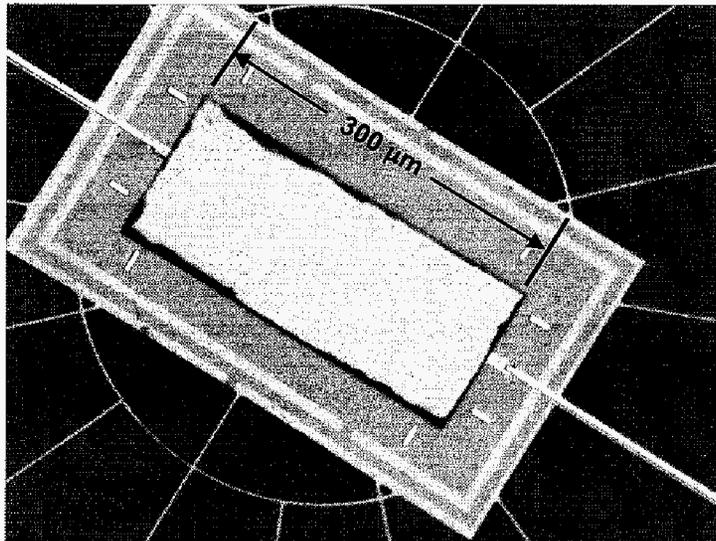
C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Optical Microscopy



Top View: Center-island device
Magnification: 100X



Center chip size: $300\ \mu\text{m} \times 100\ \mu\text{m} \times 25\ \mu\text{m}$
Chip mass: $\approx 4\ \mu\text{g}$

Web OD = 3.16 mm

Lead length = 4.66 mm

Au lead thickness = $0.3\ \mu\text{m}$

Lead width = $5.3\ \mu\text{m}$

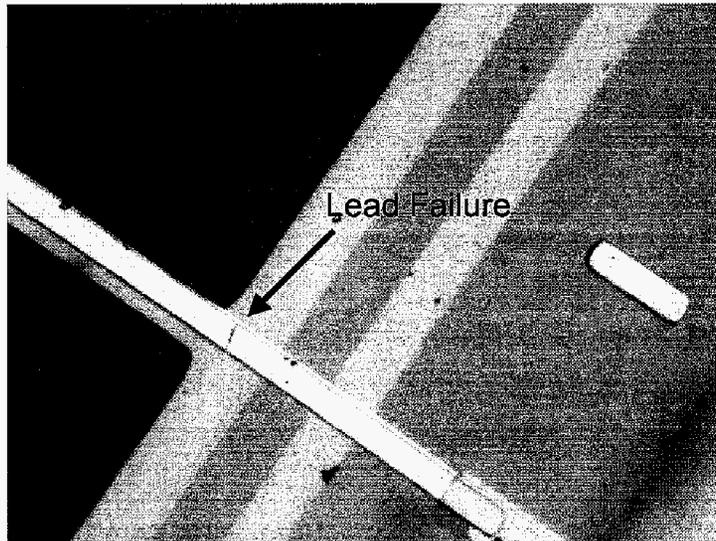
Web width = $3.4\ \mu\text{m}$

Web thickness = $1\ \mu\text{m}$

Optical Microscopy -1



Top View: Center-island device
Magnification: 500X



Open circuit condition reveals defect site in Au lead layer
Failure morphology: fracture

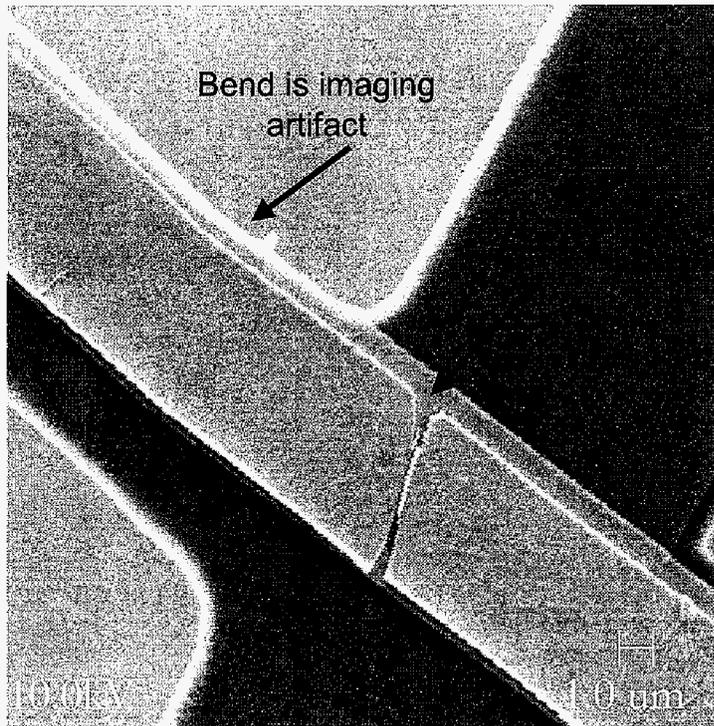
C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Scanning Electron Microscopy



Magnification: 5000X
Failure mode: tension
Fracture through upper layer
High energy fracture event



C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

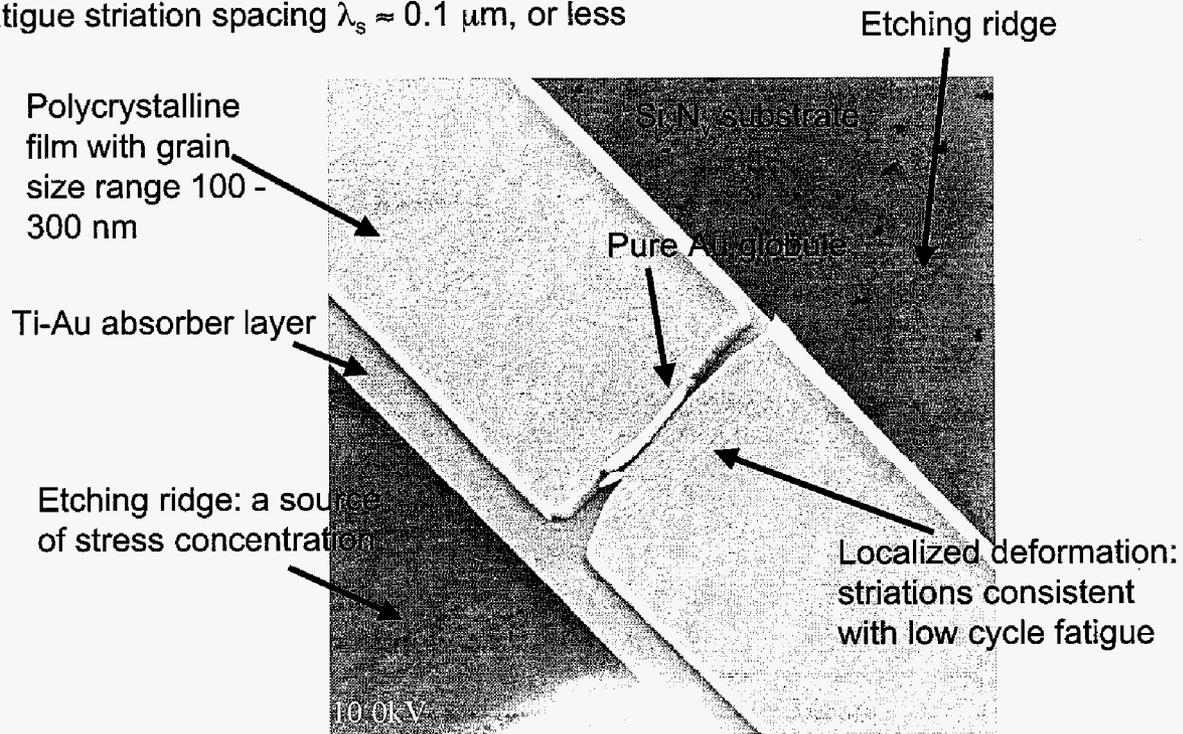
Scanning Electron Microscopy - 2



Magnification: 3000X

Plastic zone at front of crack tip, $r_y^c < \text{avg. grain size}$

Fatigue striation spacing $\lambda_s \approx 0.1 \mu\text{m}$, or less



C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

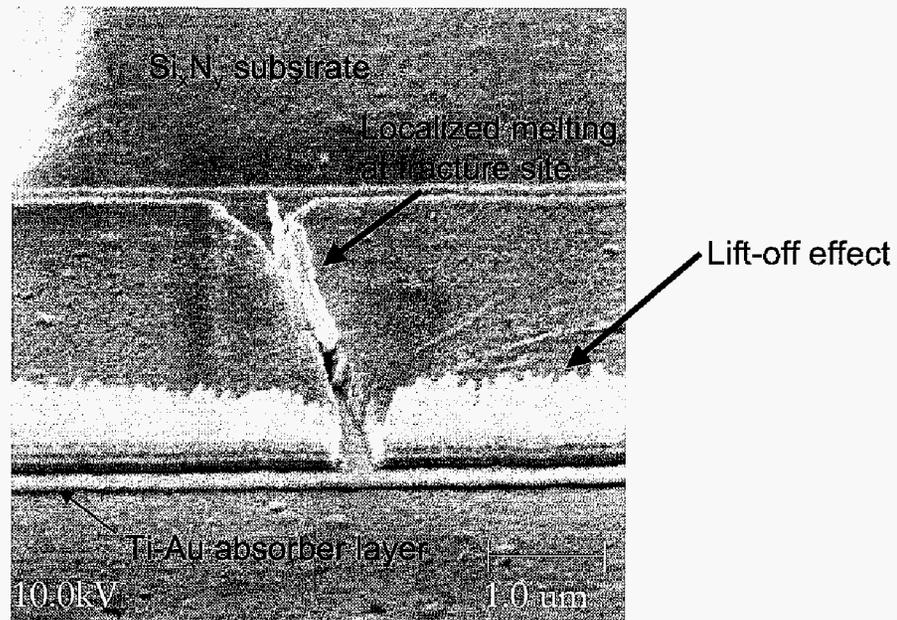
Scanning Electron Microscopy - 3



Magnification: 20,000X

Fracture completely through upper Au-layer (0.3 μm)

Localized melting at fracture site



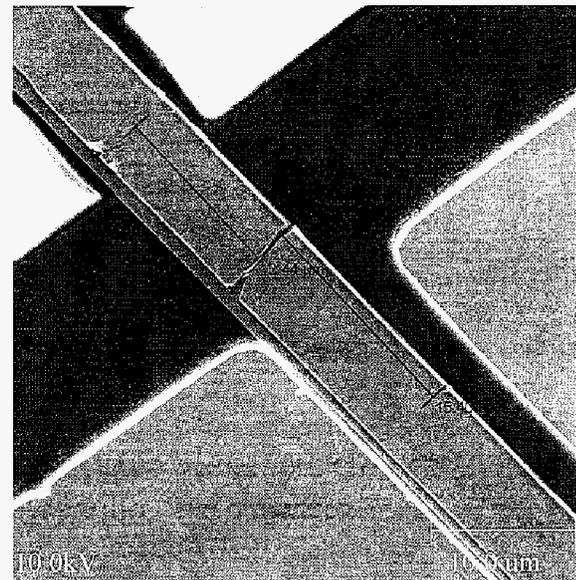
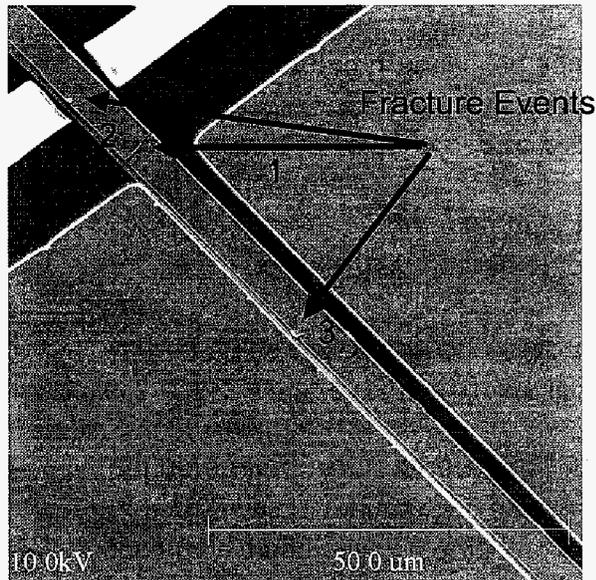
C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Scanning Electron Microscopy - 4



Magnification: 1500X
Fracture survey shows defects
distributed along lead layer, with periodicity: $\approx 10\text{-}12\ \mu\text{m}$



C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Discussion of Results



- Crystallography of slip in fcc crystals
 - Slip occurs most often on {111} octahedral planes and in <110> directions parallel to cube face diagonals
 - 12 slip systems (four {111} planes and three <110> slip directions)
- Typical crystallographic orientation in Au thin films with submicron columnar grains is <111>
 - average grain size for our films < film thickness
 - Au properties depend on orientation: $E_{\langle 111 \rangle} = 117 \text{ GPa}$, $E_{\langle 100 \rangle} = 48 \text{ GPa}$ (Courtney 1990)
- Fracture morphology
 - ductile failure, with high degree of toughness
 - fracture appears intergranular prior to rapid crack growth
 - Striations/deformation bands perpendicular to tensile axis
 - Bands formed by rotation of grains in response to applied strain
 - Multiple bands formed indicate that tensile loading was uniform
 - Au globules formed by localized adiabatic heating: plastic work converted to localized heating in deformation region
 - Au globules are not due to source spitting
 - Potential source of large adiabatic heating - dislocation avalanche mechanism of Armstrong and Grise
- Elastic limit difference between amorphous substrate and overlying film are source for large plastic response and accumulated energy

C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California

Conclusion



- Studies of the dynamic mechanical response and fatigue behavior in nanocrystalline scale Au thin films are reported
 - resonant modes of substrates calculated by hand analysis and Finite Element Modeling (FEM)
 - defect spacings were also found to correlate with resonant modes of the Si_xN_y substrate web
 - SEM images reveal that the metallic films exhibit failure morphologies consistent with low-cycle fatigue
 - close examination of the failed regions show classic fatigue striations

- Results show correlation between nanoscale microstructure and mechanical response
 - ductile failures observed at sub-micron length scales with high degree of toughness
 - localized adiabatic heating is pronounced resulting in localized melting of Au film

- Origins of these observations can be traced, in part, to the large elastic limit differences between the amorphous Si_xN_y substrate, and the over-lying metal film

Acknowledgements



- The authors would like to thank NASA and California Institute Technology
- This work was conducted under....

C. C. Hays, MRS Talk
11-29-05

Jet Propulsion Laboratory
California Institute of Technology
Pasadena, California